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**APPLICATION
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LETTERS PATENT**

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**FOR: METHOD AND STRUCTURE FOR
ULTRA-HIGH DENSITY, HIGH DATA
RATE FERROELECTRIC STORAGE
DISK TECHNOLOGY USING
STABILIZATION BY A SURFACE
CONDUCTING LAYER**

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**METHOD AND STRUCTURE FOR ULTRA-HIGH
DENSITY, HIGH DATA RATE FERROELECTRIC STORAGE DISK
TECHNOLOGY USING STABILIZATION BY A SURFACE
CONDUCTING LAYER**

5

DESCRIPTION

BACKGROUND OF THE INVENTION

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Field of the Invention

The present invention generally relates to ferroelectric storage media, and more specifically, a ferroelectric storage medium including a thin surface layer having the characteristic that charge migration is much more rapid than the charge migration characteristic of the underlying ferroelectric layer.

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Description of the Related Art

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Storage density of all magnetic disk storage devices is currently increasing at a very high rate superior even to Moore's Law. Application strategies assume this continued growth. For example, the current "computing on demand" business strategies will require very large storage databases. At the opposite end of the scale, the continued development of hand held devices requires the availability of increasingly high density miniaturized storage systems.

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Despite strategies based on the ever-increasing availability of higher density storage, the technical reality may be that magnetic disk storage technology is approaching limitations to further scaling improvement.

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Magnetic disk storage density has been increasing faster than Moore's law, at an approximate factor of double per year, but scaling of magnetic storage engineering is showing signs of breakdown. Introduction of new techniques such as AFC (Antiferromagnetic Compensation layers) has been required to maintain scaling up to now, and further projected increases in storage density rely on the new technique of perpendicular recording.

However, perpendicular recording is a complex technology which has not yet been demonstrated successfully at the required high densities. It is questionable whether magnetic disk technology can be relied upon to increase storage density beyond approximately $100\text{--}200\text{ Gb in}^{-2}$ (Gigabits per square inch).

Replacing magnetic storage media on the disk with ferroelectric media has the potential for much higher densities, since ferroelectric bits are on the scale of one nanometer. But, while magnetic dipoles cannot be depolarized by the nonexistent magnetic monopoles, the ferroelectric dipole can be. Without loss of its bulk polarization state, the surface charge can be depolarized even by stray electric charges, posing serious problems for ferroelectric disk technology. So far, no solution to this surface depolarization problem is known for ferroelectric disk technology.

SUMMARY OF THE INVENTION

Therefore, in view of the foregoing problems, drawbacks, and disadvantages of the conventional systems, it is an exemplary purpose of the present invention to provide a structure and method for solving a surface depolarization problem for ferroelectric

storage media, thereby ensuring reliable accessibility of the information stored therein in the form of polarized domains.

It is another purpose of the present invention to provide materials and thickness of materials that can be used in solving the above and other exemplary problem.

5 It is yet another purpose of the present invention to describe a read head appropriate for the inventive method.

It is yet another purpose of the present invention to describe a two-dimensional read sensor that will improve reliability and speed for the read operation of a ferroelectric storage media structured in accordance with the principles described herein.

10 To achieve these objects and aspects, as an exemplary first aspect of the present invention, described herein is a ferroelectric storage medium including a metallic underlayer, a ferroelectric data layer on top of the metallic underlayer, and a layer on top of the ferroelectric data layer having a charge migration rate faster than a charge migration rate of the ferroelectric data layer.

15 In a second exemplary aspect of the present invention, also described herein is a ferroelectric storage apparatus including a rotating support mechanism to support and rotate a ferroelectric storage medium, wherein the ferroelectric storage medium comprises a metallic underlayer, a ferroelectric data layer on top of the metallic underlayer, and a conducting layer on the ferroelectric layer. The apparatus also includes
20 read head including an electrometric sensor and a write head.

In a third exemplary aspect of the present invention, also described herein is a method of manufacturing a ferroelectric storage medium, including applying a layer of metal on a carrier substrate to serve as a metallic underlayer. A layer of ferroelectric

material is applied on top of the metallic underlayer, and a layer of conducting material is applied on top of the ferroelectric layer.

The present invention provides a solution to surface depolarization on ferroelectric storage surfaces, thereby permitting reliable ultra-high density storage.

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BRIEF DESCRIPTION OF THE DRAWINGS

The foregoing and other objects, aspects and advantages will be better understood from the following detailed description of an exemplary embodiment of the invention

10 with reference to the drawings, in which:

Figure 1 shows a conventional ferroelectric storage medium 100;

Figure 2 illustrates the extraneous charge problem 200 and an exemplary solution disclosed in the present invention to this problem by using a conductive surface layer 204;

15 Figure 3 shows models of potential 300, 310, 320 used to further explain the present invention;

Figure 4 shows an equivalent circuit 400 useful for explaining the present invention;

20 Figure 5 shows an exemplary outline of the device 500 resulting from the method taught in the present invention;

Figure 6 shows an exemplary graph 600 of the potential at the CL/air interface for the case of SBT; and

Figure 7 shows a structure 700 including an exemplary electrometer open-gate FinFET sensor 701 and a ferroelectric storage medium 702, in another exemplary aspect of the present invention.

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DETAILED DESCRIPTION OF EXEMPLARY EMBODIMENTS OF THE INVENTION

Referring now to the drawings, and more particularly to Figures 1-7, exemplary embodiments that teach exemplary aspects of the present invention will now be
10 described.

Described herein is a more detailed explanation for the above-mentioned surface depolarization problem associated with ferroelectric storage and a solution to this and other problems. That is, described herein is a method to make, and theoretically model, a disk based storage technology capable of 3000 Gb/in² (Gigabits per square inch) storage
15 density. This density would be capable of addressing the demand for ultra-high density storage.

Moreover, the read/write data rate of the design also is extremely high, approaching 10¹¹b/ s (bits per second), which is essential for several purposes. In particular, for example, whole-disk operations such as virus scan can be implemented in
20 reasonable time, particularly when the present invention is implemented using the multiple sensor array to be described later.

Ferroelectricity is intrinsically a storage phenomenon capable of an enormous dynamic range of scaling, because energies are large and local in origin, with nanoscale

domain walls. A switch from magnetic disk to ferroelectric (FE) disk storage could potentially provide the solution to continuing storage density scaling.

Because much of the materials technology required for FE disk storage can be taken over from FeRAM technology, the time and expense involved in research are much
5 reduced for FE disk storage. In addition, many of the off-disk components can be carried over from the existing storage technology.

In Figure 1, the storage in the exemplary FE disk 100 includes a layer 101 of FE material, deposited on a conducting substrate (MUL = metallic underlayer) 102. The electric polarization in the FE is directed up or down relative to the disk plane, and stores
10 a bit denoted "1" or "0" according to the sign of the polarization (the polarization pattern may be encoded relative to the actual stored information).

The polarization is written with an electric field normal to the disk surface. The polarization of the FE film is terminated at the surface by a 'bound charge' of positive or negative sign 103, 104 according to the sign of the polarization. The bit can be read
15 out by with a sensitive high speed electrometric preamplifier, which detects the sign of the bound charge, and thus the bit polarity.

United States Patent 6,515,957 to Newns et al., incorporated herein by reference, describes in more detail this broad concept.

Initial studies have successfully demonstrated the capability to electrically write
20 patterns on FE surfaces on the scale of 1000 Å or less. But a major problem emerged in the observation of a slow surface depolarization of the polarization-written information.

The problem which emerged in the initial experimental studies of writing information by vertical polarization of a FE surface, was the slow loss of surface

polarization over a few-hour to 24-hour time scale. This effect is not a loss of bulk polarization in the FE film, but was found to be due to the accumulation of mobile surface charges which neutralize the bound charges constituting the surface polarization, as illustrated conceptually in the top panel of Figure 2.

5 The slow surface depolarization phenomenon poses a severe problem for the FE disk. That is, the extended time scale of the ill-defined depolarization process means that the state of the surface is often an indefinite one, between charged and neutral. Hence, the read process cannot be engineered predictively, since prior to surface depolarization the surface is charged, while after surface depolarization the surface charge distribution
10 has been replaced by a surface dipole layer.

The top panel of Figure 2 shows a ferroelectric storage medium 200 having MUL (Metallic Underlayer) 201 and FE DL (Ferroelectric Data Layer) 202, and bound charges associated with 'up' polarized domains (+) 203 and with 'down' polarized domains (-) 204. Also shown in this upper panel are extraneous charges, from the MUL 205 or from
15 gas phase 206, that neutralize the bound charges 203, 204.

It should be noted that these extraneous charges do not destroy the information in the FE DL 202, but they do compromise the ability to read the polarized domains in the FE DL 202.

The present invention describes a solution to deal with the above-described
20 surface depolarizing charge problem. The lower panel of Figure 2 shows ferroelectric storage medium 210 similar to that of the upper panel.

If, as shown in the lower panel of Figure 2, a grounded conducting layer (CL) 211 is deliberately introduced on top of the FE layer 202, then the bound charges are neutralized in a well-defined manner by conduction within the CL.

That is, assuming the CL 211 is a good conductor, the surface depolarizing charge migration will occur very rapidly. The charge state of the surface is then determined at all times. Stated slightly differently, by providing a surface in which charges can migrate rapidly relative to the above-described time of several-to-24 hours, the present invention provides stability to the FE storage device.

The surface charge distribution can be engineered to produce a well-defined potential $\pm V_0$, just outside the CL surface, the sign depending on the sense of the FE local polarization, whose magnitude is suitable as a read input to an electrometric sensor as described in the next section.

By fixing the problem of slow surface charge depolarization, the ferroelectric storage medium solution becomes viable, provided it can be demonstrated that:

1. The read process is viable when the surface charge distribution is a dipole layer;
2. The CL can indeed redistribute the neutralizing charges rapidly enough;
3. A satisfactory voltage swing $\pm V_0$ is obtained for the read sensor with realistic materials and dimensions;
4. The write process remains viable;
5. A practical read sensor can be designed; and
6. The data transfer rate in read and write is adequate.

In the following, and in more detail in D.M. Newns, “Ultra-High Density Ferroelectric Storage Medium Stabilized by a Surface Conducting Layer”, IBM Report, 07/01/2003, it is demonstrated that these above-identified six issues can be satisfactorily dealt with in a practical design.

5 ELECTROMETRIC READ WITH DIPOLE LAYER

First the qualitative issue of electrometric readout from a surface dipole layer is addressed. In the bottom panel of Figure 2, the disk surface, including the FE polarization, the CL charge, and their images in the MUL, is always neutral (it would be so even in the absence of the CL, though the CL reduces the strength of the dipole layer significantly).

Electrostatically, the polarized FE layer presents a dipole layer, across which there is a voltage $\pm V_0$, the sign depending on the sense of the FE polarization, between air just outside the medium surface and the grounded MUL. But there is no field normal to the surface in air, so it may at first seem that there is no electric field source for the read sensor.

Hereinbelow, it is explained, from simple electrostatics, how electrometric read is possible under these conditions. Essentially, when the read plate is put in position, the charges rearrange themselves to minimize energy, resulting in a field which charges the read amplifier input capacitance.

Referring to Figure 3, a simple model is presented to show how this process works. The top panel 300 of Figure 3 shows dipole layer 301 having strength V_0 atop grounded metal underlayer (ML) 302. Outside the dipole layer 301, there is no field, but the potential is V_0 .

The middle panel 310 shows an isolated metal plate 311 with capacitance C_g 312 to ground. The bottom panel 320 shows the metal plate 311 located atop dipole layer 301, thereby generating a field E 321 and charge Q_g 322 on the capacitor 312.

That is, in Figure 3, the FE+CL is represented as a dipole layer 301 of strength V_0 . In the absence of a read plate the dipole layer atop the conducting, grounded MUL produces a potential V_0 in air, with no field (e.g., the top panel 300 in Figure 3).

The read capacitance is an isolated metal plate P 311, at $x = d$, connected via a capacitor C_g 312 to ground. In the absence of a dipole layer 301, the potential is zero everywhere (e.g., the center panel 310 in Figure 3).

Now, the dipole layer and read plate are introduced together (e.g., the bottom panel 320 in Figure 3). As developed in the above-referenced IBM Report dated 07/01/2003, the voltage on the read sensor 311 is:

$$V_g = V_0 / (1 + C_g / C_p), \quad (1)$$

where the plate capacitance C_p is defined as

$$C_p = \frac{A}{4\pi(a+d)}, \quad (2)$$

The electric field and capacitor voltage have become nonzero (a similar situation occurs in the well-known Kelvin probe).

The read signal can be modeled by the equivalent circuit 400 in Figure 4, and can be interpreted as the dipole layer, acting like a battery 401, in series with the potential divider formed by the two capacitances C_p 402 and C_g 403. The capacitance C_p 402 is much smaller than C_g 403, which results in a significant reduction in read sensor input (e.g., see Equation (1) above, and the description below).

The voltage V_g 404 will be the input voltage on an FET (not shown in Figure 4), whose gate capacitance is C_g 403. In its practical realization, the floating gate is eliminated from the read sensor design, to give direct coupling between the FE medium and the sensor FET channel, as seen below.

5 Structure 500 of the FE disk design is shown in Figure 5. The active part of the disk includes a Metallic UnderLayer, MUL 501, the ferroelectric Data Layer 502, and a top coating, the Conducting Layer CL 503. The MUL 501 may be formed of a metal or other conducting medium. For example, 30 Å of SrRuO_3 was used as the MUL in the structure described in Ahn et al., Science 276, 1100, 1997. A substrate 510 supports the
10 above structure.

The DL 502 preferably is a thin layer (e.g., on the order of tens of angstroms) of the insulating ferroelectric data storage material, with FE polarization P . The CL 503 preferably is a thin layer of conducting material, such as amorphous Si or a doped perovskite conductor. A thickness preferably is within a range from about 5 Å to about
15 25 Å, and more preferably about 15 Å. The MUL 501 and CL 503 are electrically grounded.

The read head 504 preferably is an electrometer sensor including an FET input stage 505, which is the preamp for a read amplifier 506. The write head 507 includes a write plate 508 connected to the output of a write amplifier 509.

20 In operation, the write plate 508, driven by the write amplifier 509, forms one plate of a capacitor, with the other plate being the grounded MUL. The electric field normal to the disk produced by the write plate 508 polarizes the FE layer 502 beneath the plate 508, normal to the surface, and in the sense of the field.

The read sensor 504 picks up variations in potential present due to the difference in potential above the 'up' and 'down' polarized bits, and amplifies them, enabling readout of the sense (e.g., "0" or "1") of the bit.

The FE 502 preferably is a good insulator, is polarizable by the write field, and is capable of response within the required time frame.

The CL 503 preferably is a good conductor, and has an appropriate charge profile and energy level structure to enable efficient read and write operation.

An exemplary test case is now described for analysis. This must be an aggressive one, and a 3000 Gb/in² storage medium is analyzed here and in more detail in other specifications. In Table I, the first two columns give the bit dimensions, a and b refer to the thickness of the FE 502 and CL 503 layers respectively, d is the fly height between the top of the CL 503 and the bottom of the R/W head 504,507, and v_{disk} is the disk velocity:

Table I: Test Case Parameters

$\parallel \text{Track } (\text{\AA})$	$\perp \text{Track } (\text{\AA})$	$a (\text{\AA})$	$b (\text{\AA})$	$d (\text{\AA})$	$v_{\text{disk}} (\text{ms}^{-1})$	$\tau_{\text{R/W}} (\text{ns})$
100	200	100	~ 15	30	40	0.3

The read/write time $\tau_{\text{R/W}}$ follows from the bit length and disk velocity and is a key input.

The MUL layer 501

As described in the above-cited article by Ahn et al., 30 \AA of SrRuO_3 was used as the MUL 501, deposited onto an STO (SrTiO_3) substrate. The MUL 501 is not considered critical as regards its electronic properties. However, its material's properties are key as a substrate for the FE layer 502.

The FE layer 502

Table II shows a set of FE materials which have been proposed in the FeRAM field. This area has very similar materials requirements to those of FEdisk technology.

Listed in the table are FE polarization P , coercive field E_c , and FE dielectric constant ϵ_{ox} .

Table II: Ferroelectric Materials

Material	P (μCcm^{-2})	E_c (KV/cm)	ϵ_{ox}
PZT ($\text{Pb}(\text{Zr}_x \text{Ti}_{1-x})\text{O}_3$)	~ 50	100 – 200	40 – 2000
SBT ($\text{SrBi}_2\text{Ta}_2\text{O}_9$)	6, 14	54	124, ~ 200
BaMgF ₄	3 – 8	?	?
STN ($\text{Sr}_2(\text{Ta}_{1-x}\text{Nb}_x)_2\text{O}_7$)	0.4	30	55
NFM (COVA)	0.2	50	7 – 12

Results for any material depend on preparation, and in the cases of variable composition, also on composition. They are listed in approximate order of the FE polarization P , which varies over more than two orders of magnitude. The listing also turns out to be approximately the order of their dielectric constants ϵ_{ox} .

In contrast, the critical field E_c does not vary very much from material to material (though the PZT value is significantly larger, making it more stable). The last line of the Table is a proprietary material developed by the COVA Corporation, termed “NFM” (New Ferroelectric Material). Experimental data on PZT show that response times can be sufficiently short relative to $\tau_{R/W}$. Similar data on the other materials in Table II is not currently available.

Modeling studies during development of the present invention have focused on the materials SBT and STN, with polarization lying in the intermediate range.

PZT can also be used as the FE material, relative to SBT and STN, its higher polarization relative to E_c leads to a stronger tendency to depolarize, and to a requirement

for relatively higher carrier concentration in the CL. Its read signal is relatively large, and, similarly, a larger write field is required relative to SBT and STN.

The NFM material leads, on the contrary, to a low requirement on CL carrier concentration, and to lower read signals and write fields relative to SBT and STN.

5 The Conductive Layer CL

The CL preferably is a good conductor and enables field penetration normal to the disk plane. That is, it cannot completely shield the FE from the read and write plates.

1. CL Conductance

10 The resistance R per square times the capacitance C of the bit defines a time constant for lateral charge dissipation in the CL. This preferably is shorter than 10^{-10} seconds to enable charge response within the read/write time $\tau_{R/W}$. Hence, one obtains the condition:

$$RC < 10^{-10} \text{ s.} \quad (3)$$

15 The carrier density in the CL can be assumed to be at least equal to the FE polarization P . Hence, inserting $R = 1/(P\mu)$, where μ is the mobility of the carriers in the CL, a constraint on the mobility is:

$$\mu < 10^{10} \frac{C}{P} = 0.03 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}. \quad (4)$$

20 Here $C = \epsilon_{\text{ox}} \epsilon_0 A/a$, where A is bit area, and assume $\epsilon_{\text{ox}} = 1000$ for PZT, giving a close to maximal value for the required CL mobility. The requirement on the mobility is seen not to be severe. It can be satisfied by amorphous Si or by a doped perovskite conductor.

A perovskite, in its simplest pure form, is a material of formula ABO_3 , where B is a transition metal and A is a divalent metal (e.g., SrZrO_3).

2. Shielding by CL of the FE Medium from Read/Write Heads

Shielding by the grounded CL inhibits sensing of the FE polarization by the read head, and also penetration of external write fields into the FE medium. Shielding places
 5 severe constraints on the CL, decreasing the available read voltage, and increasing the required write voltage.

A measure of the effectiveness of shielding is the Fermi Thomas screening length λ^{-1} in the layer versus the CL layer thickness b , where λ is given by

$$10 \quad \lambda^2 = \frac{4m^*}{r_B \epsilon} \left[\frac{3n_{3D}}{\pi} \right]^{1/3}, \quad (5)$$

m^* being effective mass, ϵ is the CL background dielectric constant, n_{3D} is the carrier concentration in the CL, and r_B is the Bohr radius. If

$$\lambda b \gg 1 \quad (6),$$

15 then shielding is highly efficient, which is an undesirable result. However, for reasonable values of CL carrier concentration and dielectric constant, λ^{-1} can be made several Å, and for layers of thickness $b \approx 15 \text{ Å}$, the shielding can be made acceptable. The shielding effect will limit CL layer thickness to below approximately 20 Å, depending on CL material.

20 Particular cases will be investigated in the following.

It will typically be assumed that the aerial carrier concentration in the CL is $N = 2P/e$. This ensures that, as a result of screening the FE polarization P , the carrier concentration is bounded between the limits P/e and $3P/e$, for bits of depletive/
 accumulative sign respectively, remaining not only nonzero, but allowing close to linear
 25 screening of the FE polarization by the CL to be maintained.

Two practical choices for the CL are particularly attractive. One choice is Si, and the other is doped perovskite or similar oxide.

3. Silicon CL

5 There is extensive experience in the FeRAM field with the FE/Si interface. The doping in a Si layer can be readily controlled. The required doping is, however, extremely heavy. A 15Å layer with $N = 2P/e$ requires $2.5 \times 10^{21} \text{ cm}^{-3}$ for PZT and $3 \times 10^{20} \text{ cm}^{-3}$ for SBT.

10 The layer is exemplarily deposited as amorphous Si. The mobility is adequate to satisfy condition (4) for this application. An advantage of Si is that it can readily be given a thin layer of SiO₂ as protection. Additionally, its parameters are well known.

For degenerate p-type Si, the formula for the FT screening length is

$$\lambda^2 = \frac{4\sqrt{2}}{\pi \epsilon} E_F^{1/2} (m_{lh}^{3/2} + m_{hh}^{3/2}) \quad , (7)$$

15 where E_F is Fermi energy w.r.t. VB edge, and m_{lh} and m_{hh} are the light and heavy hole masses.

20 The shielding effect was modeled in detail. The results herein are based on linear screening theory, for a CL thickness of $b = 15\text{\AA}$ (giving 12 layers of Si atoms), doping is p-type, the carrier density has been chosen so that the aerial carrier density is twice the polarization of the FE, thus ensuring that the FE polarization can be neutrally screened for either bit polarization.

Results are shown in Table III for SBT and STN. The Table includes the assumed values of Polarization P , and FE dielectric constant ϵ_{ox} . The read potential is given in the form of the transition potential ΔV_R , the difference between the potential on the top

surface of the CL for a positively-upward FE polarization, and that for a negatively-upward polarization.

The Table also includes the write voltage at the write plate required to generate the critical field E_c in the FE medium.

5 Table III: Results for 15Å Si CL

FE	P(e /cell)	ϵ_{ox}	ΔV_R (mV)	V_W (V)
SBT	0.06	150	56	34
STN	0.004	50	20	5.5

The transition potential ΔV_R at the CL surface is in the tens of millivolt range. It is undesirably small due to CL shielding, but is detectable by a specialized electrometer sensor head described below. The write voltage is of order tens of volts and seems practical. The write issue is not further discussed, as it is non-critical to the present invention.

4. Transition Potential

For the same p-type Si model, one can calculate the transition potential $V(x)$ at the CL surface at the interface between a positively polarized and a negatively polarized bit. Again using a linearized Fermi Thomas model, the potential is illustrated in Figure 6. Figure 6 shows the potential at the CL/air interface for the SBT case, with the vertical axis in volts, and the horizontal axis in units of 50 Å.

As shown, the transition occurs within one half a bit length on each side of the transition. The transition magnitude is somewhat less than for the uniformly polarized surface, being reduced from 56 mV to 38 mV, but this is still a workable transition.

5. Doped perovskite CL

An alternative to heavily doped Si is a doped perovskite or similar oxide with compatibility to the FE surface, or even a few doped surface layers of the FE itself.

The required doping levels of up to $3 \times 10^{20} - 3 \times 10^{21} \text{ cm}^{-3}$ are normal for perovskite semimetals. The mobility is adequate to satisfy condition (4) above. In the above-cited Ahn article, a 30 Å layer of SrRuO₃ was described as successfully deposited with 1.5 Å roughness over a $6 \times 6 \text{ } \mu\text{m}^2$ area. Therefore, depositing one half of this thickness of perovskite, as required, is easily attainable.

The parameters for perovskite are less well known than for silicon, a disadvantage in modeling. Consider a layer of doped perovskite located (epitaxially, if the substrate/CL lattice match allows it) atop the FE. The thickness is typically $b = 16 \text{ Å}$, *i.e.*, $n = 4$ unit cells of perovskite. As a benchmark, 30 Å of SrRuO₃ was successfully deposited in the technique described in the Ahn article, with 1.5 Å roughness.

The perovskite CL is assumed to be n-doped, when the Fermi level lies in the perovskite transition metal d-band (e.g., at the bottom of the 3d band for STO). Since the transition metal forms a simple cubic lattice with a narrow band, it is modeled by tight-binding theory, taking into account the finiteness of the film by eliminating the hopping integrals from the first layer of the film to the non-existent zeroth layer, and from the n th layer to the non-existent $(n + 1)^{\text{th}}$ layer.

In the in-plane directions, the layer is taken as infinite. As before, the aerial doping of the CL is taken as two times the FE polarization. The modeling details are further outlined in “Ultra-High Density Ferroelectric Storage Medium Stabilized by a Surface Conducting Layer”, D.M. Newns, IBM Report, 07/01/2003.

The dielectric constant of the CL is ϵ , the band mass is m , and a simple s -type band structure is assumed for simplicity. Results are shown in Table IV.

Table IV: Results for $n = 4$ Perovskite CL Model

m/m_e	ϵ	ϵ_{ox}	$P (e /\text{cell})$	$V(0) \text{ (mV)}$	$\Delta V \text{ (mV)}$
1	20	150	-0.06	-23.5	49.5
1	20	150	0.06	26.0	

This model is not linearized, so the potential $V(0)$ at the surface of the CL depends somewhat on the sign of the polarization. The transition potential is again 50 mV, about the same value as in the quite different Si calculation for a similar layer thickness.

THE ELECTROMETRIC SENSOR

FinFET Input Stage

A conceptually simple exemplary design for the electrometric sensor is based on some radical modifications to the 20 nm channel length FinFET described recently by Doyle, et al., “Transistor Element for 30nm Physical Gate Length and Beyond”, Intel Technology Journal, Vol. 6, Issue 2, Section 5 of 9, published May 16, 2002 (available on the Internet at http://www.intel.com/technology/itj/2002/volume06issue02/art05_transistorarch/p05_architecture.htm).

Referring back to the read head 504 in Figure 5, on one side of the Si fin a standard top gate is located, mainly for bias purposes. The opposite side of the channel is gate-free, with the CL as the effective (but weakly coupled) gate carrying the high frequency input signal. The axis of the 20 nm channel (e.g., the source/drain axis), is oriented transverse to the track, so as to match the 20 nm bit width. The width of the

FinFET channel is approximately 10 nm, and is designed to match the bit length (the described design had a 13 nm wide channel).

An analysis of this sensor, which acts as the read input preamplifier, is provided in the above-cited IBM report by Newns. The output voltage is about 0.1 V, which
5 greatly exceeds Johnson noise, and the output impedance is adequate for the next amplification stage, given the required bandwidth.

The amplifier should contain a high-pass filter to remove any low-frequency sources of noise ($1/f$ noise). An additional source of noise will be charge fluctuations in or on the sensor oxide, which sometimes may be fast events, such as sporadic tunneling
10 events, not eliminated by the high pass filter.

If these rapid charge fluctuation events occur at a low enough rate (e.g., such as below one in 10^5 bit reads) to be captured by the decoder error correction, they will have no effect. If not, hardware fixes exist at some increased complexity (see below).

Multiple FinFET Sensor Array

15 There are advantages to using a multiple sensor array, which can be readily manufactured by the same lithographic means as the individual FinFET sensor. Sensors, and associated write heads, can be placed on parallel tracks, in order to increase the speed of data transfer (an important consideration in ultra high density storage). Different along-track mutual displacement of the bits and sensors in different tracks will be
20 introduced to reduce crosstalk.

Secondly, multiple sensors can be placed along the same track. By combining the digitized output from each along-track sensor with an appropriate delay, and polling the resulting set of signals, a significant reduction in signal/noise can be obtained. In this

way, the effect of relatively high occurrence rate rapid charge fluctuation events associated with the sensor oxide can be eliminated.

A ten parallel track R/W head will have a data rate of 0.4×10^{11} b/s, enabling a 3 in. disk to be written and read in about 20 minutes, which should be adequate.

5 In sum, the FE disk is an attractive candidate for an ultra high density storage medium. However, FE media seemed inoperable due to the existence of a slow surface charge depolarization problem.

The present invention provides a solution to the depolarization problem, which is, for example, in the exemplary embodiment, to deposit on the surface of the FE disk a
10 high mobility conducting layer, thus placing the FE in equilibrium with a uniform chemical potential. Hence, slow electronic charge rearrangements have no driving force. The layer also stabilizes the bulk polarization in the FE, by reducing depolarizing fields.

If the CL is made reasonably thin (e.g., within a range of about 5-25 Å), and of appropriate material (e.g., amorphous Si or a doped perovskite conductor), the electric
15 field seen by an external electrode is sufficient to enable read and write functionality of the bit, for both Si and perovskite conducting layers.

The read process involves electrometrically sensing the transition potential between oppositely polarized bits, which is found to be adequately sharp. The read sensor analyzed above involves a FinFET with exposed sides and top, the CL forming a weakly-
20 coupled bottom gate.

A simple preamplifier with adequate signal/noise and output impedance is then formed by the sensor and output resistor. A degree of uncertainty is provided by the

possibilities for charge noise in/on the exposed gate oxide. The noise from this process can be greatly ameliorated in a sensor array design.

The write process involves a simple write plate at some tens of volts, and is considered non critical. Multiple R/W channels can be lithographed in the sensor array design, enabling R/W data rates approaching 10^{11} bits per second.

Thus, the present invention maintains surface charge stability in the FE disk medium, while maintaining read and write capability.

While the invention has been described in terms of an exemplary embodiment, those skilled in the art will recognize that the invention can be practiced with modification within the spirit and scope of the appended claims.

Further, it is noted that, Applicant's intent is to encompass equivalents of all claim elements, even if amended later during prosecution.